## Phonon Filtering by Nanovoids In Silicon: a Computational and Experimental Analysis

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### Background

The role played by morphological defects (md's) on thermal conductivity and phonon MFPs has been the subject of a wealth of papers recently appeared in literature.



Spectral Phonon Transport Properties of Silicon Based on Molecular Dynamics Simulations and Lattice Dynamics

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### Aim of this work

- Investigate the interplay among md size and density
- Compare thermal conductivity trends when fixed-size md's are also present
- Reference system is silicon, both single- and nano-crystalline
- Fixed-size md's are grain boundaries (GBs)
- Variable-size md's are nanovoids (NVs)

### Helium implantation

Heat treatment



#### Nanovoids generated by Heimplantation in scSi



G. F. Cerofolini et al., Mater. Sci. Eng., R; 27 (2000) 1-52

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### Sample preparation & TDTR

Double I<sup>2</sup> on 450-nm thick nanocrystalline Si films grown by CVD on oxidized Si 1. 90 keV,  $4 \times 10^{16}$  cm<sup>-2</sup> 2. 58 keV,  $1.5 \times 10^{16}$  cm<sup>-2</sup>







### NV Shape and Distribution

#### Sample B6S1: Annealed at 500°C



#### WITHIN THE GRAINS: Void diameter $d_p \approx 2 - 4$ nm Void density $\approx 3 \times 10^{17}$ cm<sup>-3</sup> Void spacing $d_s \approx 11$ nm

#### Sample B6S6: After the whole cycle



## Coalescence of voids as in the single-crystal case

B. Lorenzi, et al.; J. Electron. Mater., 43 (2014) 3852-3856

### NV Shape and Distribution

#### Sample B6S1: Annealed at 500°C



WITHIN THE GRAINS: Void diameter  $d_p \approx 2 - 4$  nm Void density  $\approx 3 \times 10^{17}$  cm<sup>-3</sup> Void spacing  $d_s \approx 11$  nm

#### Sample B1S1: Annealed at 1000°C



Void diameter  $d_p \approx 1 - 2 \text{ nm}$ Void density  $\approx 2 \times 10^{18} \text{ cm}^{-3}$ Void spacing  $d_s \approx 6 - 9 \text{ nm}$ 

B. Lorenzi, et al.; J. Electron. Mater., 43 (2014) 3852-3856



A. S. Henry & G. Chen, J. Comput. Theor. Nanosci. 5 (2008) 1–12. A. Eucken, Ceram. Abstr. 11 (1932) 576

J. Tang et al., Nano Lett., 10 (2010) 4279-4283

### A possible explanation

$$\varphi \equiv \frac{V_{\rm p}}{V_{\rm tot}} = \frac{4\pi}{3} \frac{\left(d_{\rm p}/2\right)^3}{\beta (d_{\rm s} - d_{\rm p})^3}$$





If porosity  $\varphi$  is constant,  $d_{\rm s} \propto d_{\rm p}$ 

Thus, the spacing  $d_s$  between scattering centers scale with the *roughness*  $d_p$  of the scattering surface.

- *d*<sub>s</sub> sets the MFP of scattered phonons
- *d*<sub>p</sub> sets the wavelength range of scattered phonons



### **Concurrent scattering at NVs and GBs**

- With no NV, MFP is limited by GB only, affecting phonons with  $\lambda < d_{GB} = 50$  nm
- For large  $d_s$ ,  $d_p$  ( $\approx d_{GB}$ ) NVs have little effect as they scatter phonons already scattered ٠ by GBs (B6S6 and B6S5)
- The more  $d_s$  and  $d_p$  decrease, the more the MFP of phonons with  $\lambda < d_p$  decreases (MFP) ٠  $\approx d_s$ ) (B6S3, B6S2 and B6S1), i.e. NVs decrease the MFP of phonons with  $\lambda < d_p$  to  $\approx d_s$
- For very small  $d_s$  and  $d_p$  (B1S1) NVs scatter phonons of very low  $\lambda$  (<  $d_p \approx 1$  nm), that • however marginally contribute to  $\kappa$ . Thus they are ineffective on  $\kappa$ , that gets back to its NV-free value

# Computational framework

Method: Approach-to-Equilibrium Molecular Dynamics

• AEMD solution of heat-transport equation under PBC for an initial steplike initial temperature profile

C. Melis et al., Eur. Phys. J. B. 87, 96 (2014)

• Time-dependent temperature difference between the two regions

$$\Delta T(t) = \sum_{n=1}^{\infty} C_n e^{-\alpha_n^2 \left(\frac{\kappa}{\rho c_v}\right)t}$$

where:

•  $\Delta T$  is fitted to obtain  $\kappa$ 



### **Computational framework**

#### **Investigated systems**:

Ordered porous sc-Si (OPscSi)
Random porous sc-Si (RPscSi)
Quasi-random porous nc-Si (RPncSi)

In OPscSi the porosity  $\varphi = \varphi(N_p, d_p)$  actually depends on the number of NVs  $(N_p)$  and on NV diameter  $(d_p)$ In RPscSi a uniform  $\varphi$  is generated along the sample

#### Simulation protocol:

- 1. high-temperature simulated annealing
- 2. careful equilibration at room temperature
- 3. inner surfaces fully relaxed to a highly-defected structure
- 4. used EDIP potential [Justo et al., PRB 58 (1998) 2539]



### Trends in ncSi





### Possible reasons for the discrepancy

- NV distribution in the simulation is not uniform, NV segregating close to the GBs (thus,  $d_{\rm s} \not \propto d_{\rm p}$ )
- NVs may not be fully described by their size, i.e. termination of inner surfaces by e.g. H may make ultrasmall NVs qualitatively different from standard NVs

E. Romano et al., *Surf. Interf. Anal.*, **42** (2010) 1307; E. Romano et al., *Surf. Interf. Anal.*, **42** (2010) 1321; G.F. Cerofolini et al., *Surf. Sci.*, **604** (2010) 1215

• Other ideas?

### **Summary and Conclusions**

- Ion implantation demonstrated a suitable technique to introduce controlled porosity in silicon
- NVs were found to modulate κ depending on their spacing and size
- Euken model of thermal conductivity was falsified
- AEMD simulation quantitatively confirmed the effect of NV in ncSi on  $\kappa$
- Gray model of thermal conductivity confirmed inadequate to explain the physics of the system
- A question remains open about why ultrasmall NVs do not impact on  $\kappa$

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